

Fig. 9.- External and internal layers of an aluminized low-carbon steel. x 250

ON THE BEHAVIOUR AGAINST ANODIC POLARIZATION OF AMORPHOUS ALLOY, METGLASS 2826.

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O. ABSTRACT.

The evolution of the polarization anodic curves of amorphous, Metglass 2826 alloys in SO $_4$ ²⁻, PO $_4$ ³⁻ dilute solutions, has been studied electrochemically.

These results show the ranges where they are active (fast metal dissolution) and or passive (formation of protective layers). The mechanisms and interaction parameters through the electrolyte/metal interface have been studied too.

1. EXPERIMENTAL.

Samples of metallic glass, Metglass 2826 (40% at.Fe, 38 Ni, 18 B, 4 Mo) were previously cleaned with dilute Cl₄H, and degreased with aceton. Diluted solutions of $SO_4^{\,2-}$, $PO_4^{\,3-}$ were used as electrolytes.

The acidity variations were carried out adding in a convenient way droplets of $\rm H_2\,SO_4$, $\rm H_3\,PO_4$ or NaOH. A Wenking POS-73 was used for potenciostatic measurements, with a scanning velocity of about $\rm 1mV/seg$. All measured potencials were referred to the Satured Calomel Electrode (SCE). The electrolytic solutions were neither stirred nor aereated.

2. RESULTS AND DISCUSSION.

a) EVOLUTION OF ANODIC POLARIZATION CURVES IN SO42-0.1M SOLUTIONS.

Fig. 2.1 shows the anodic evolution of Metglass in acid solutions. There is a small change of the maximum anodic current densities (J_m) with pH, fluctuating around 58mA/cm^2 value, with correspondent transicion active-passive potencials about E=800mV. The passivation current densities (J_p) increase rapidly

between 0.55 and 2.8 mA/cm², according to the disminution of pH.

These results would indicate that passive layers became more

unstable as the solution's acidity decreases.

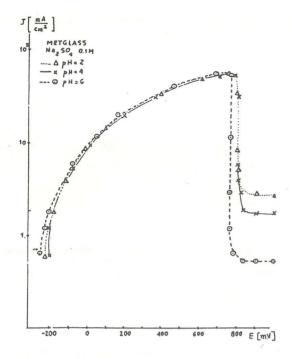


Fig. 2.1. Polarization curves of the Metglass 2826 in SQ4 2-0.1M

b) EVOLUTION OF ANODIC POLARIZATION CURVES IN DILUTED PHOSPHATE SOLUTIONS, (PO43-). Fig. 2.2.

The main anodic reactions are iron (Fe \Longrightarrow Fe²⁺ + 2e) and nickel (Ni \Longrightarrow Ni²⁺ + 2e) oxidations. Fe²⁺ suffers further oxidation to Fe³⁺.

The anodic polarization curves strongly depent on the pH. The transpassivation current density (J_{tp}) is scarcely dependent on pH variations. For basic pH increases for lower potencials (tension line of oxygen).

c) <u>DEPENDENCE OF ANODIC Ja (TO CONSTANT POTENCIAL)</u>, <u>MAXIMUM (Jm)</u> <u>AND PASSIVE (Jp) CURRENT DENSITIES ON pH, PO4 3 - 0.2M.</u> Fig. 2.3.

There is a positive $Ln(J_a)$ dependence on pH for a constant potencial (i.e. E=-200 mV). The active participation of hidroxyl ions in the anodic dissolution allows us to say that J_a obeys the Buttler-Volmer equation (1) in the form: $Ln(J_a)=\alpha pH+cte$.

The negative ${\rm Ln}(J_m)$ dependence on pH, indicates the trend of the anodic reaction to passivation with the formation of semistable products. There is a linear dependence of ${\rm Ln}(J_p)$ on pH in the acid range. The dissolution and/or formation of passive layers is controlled by the polarization potencial in the oxide(hydroxide)-solution interface.

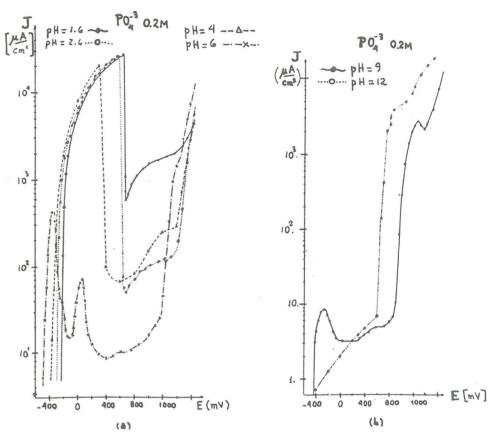


Fig. 2.2. Polarization curves of Metglass 2836 in PO 3 - 0.2M: a) Acid, b) Basic.

 J_P is nearly constant for basic pH. The oxygen evolution favours the formation of protective layers (2), whose structure are not known yet (there are different models).

Some elemental reactions of Fe in these solutions, for layer passive's formation, could be:

3 Fe₂₊ + 4 H₂O
$$\Longrightarrow$$
 Fe₃O₄ (?) + 8 H⁺ + 2e
or 2 Fe + 2 PO₄³⁻ + 2H₂O \Longrightarrow γ -Fe₂O₃ (?) + P₂O₃ + 4 OH⁻ + 2e

Moreover, Ni would be forming semi-stable complexs, particulary in the pre-passive range, with elemental reaction (3):

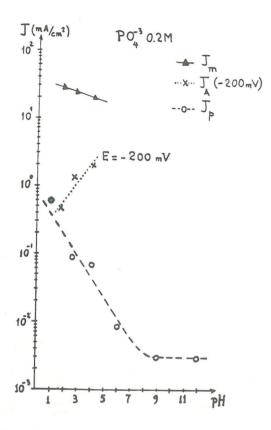


Fig.2.3. Dependence of the current densities, anodic J_1 , maximum J_1 , passivation J_2 on pH for PO 3 - 0.2M solutions.

3. CONCLUSIONS.

-Current densities maximum J_m are scarcely dependents with pH. -Current densities passive J_p became more unstable as the solution's acidity decreases. These current densities are nearly constant for basic pH.

ACKNOWLEDGEMENT: The autor wishes to express his grateful acknowledgement to Dr. Domingo Aliaga G., director of the Mössbauer-Corrosion Lab. Science Faculty, Universidad Nacional de Ingeniería, Lima-Perú.

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CORROSION IN SEA-COAST: COLLECTION AND DETERMINATION OF MARINE ATMOSPHERIC CHLORIDES

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1.INTRODUCTION

Atmospheric corrosion has been studied theoretically, in the laboratory(by simulated or accelerated tests) and in field exposure tests. These types of studies complement each other, but the last one is very important to know the atmospheric condictions to metal corrosion. Some regions of the World, as for example USA or URSS, are well studied. In Europe, some countries have also studied the atmospheric condictions to metal corrosion, but other countries, such as Spain, begin now to star work in this field.

The atmospheric corrosion of metals is greatly influenced by metereological conditions, such as humidity, temperature and its variations, wind, rainfall and dew point. These factors influence the formation of the electrolyte layer on the metal surface, which is responsible for atmospheric corrosion. In addition to the metereological conditions, the chemical composition of air have a great importance, specially the presence of chlorides and sulfur dioxide. The first one is always present in sea-coast and harbours atmospheres and the second one is present when these places are near from industrial areas which produce pollution. The chlorides arise mostly from the sea and return to the ground fundamentally by precipitation by rain and by dry fallout. Both rain and dry deposited chlorides represent an aggresive factor for metallic structures. As rain water is not reteined in the flat inclined samples used currently in atmosphericcorrosion tests sites, total chloride collected seems therefore more relevant than the amount rain-borne, in trying to correlate corrosion rates with atmospheric chlorides.

Two main methods to collect chlorides from the atmosphere have been described in the literature: the "wet candle"method (1) and the "collecting metallic trough"method(2). However there are not enough bibliograpic information to compare the

Portugaliæ Electrochimica Acta, 7 (1989) 143-148